

Bridging the gap:

Gaussian **A**pproximation **P**otential

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Outline

- from QM to interatomic potentials
- potential based directly on detailed QM data
- high dimensional fit (Gaussian Processes)
- atomic neighbourhoods: bispectrum
- the first GAP for carbon
- other uses: defining the local energy

From QM...

- quantum mechanics is the 'ultimate truth'
- expensive to solve
- sequence of approximations:
 - Full CI
 - QMC
 - DFT-LDA
 - tight binding
- interatomic potentials

...to interatomic potentials

- energy is sum of atomic energies
- atomic energy depends on neighbouring atoms
- electronic problem is not solved

- cluster expansion of total energy

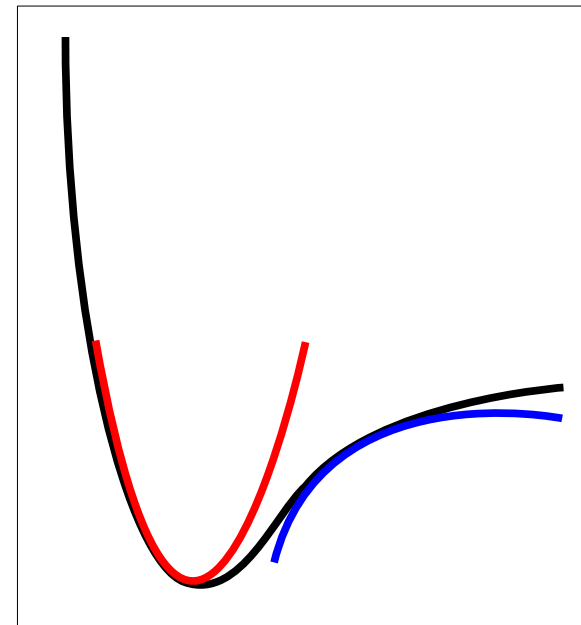
$$V(\mathbf{r}_1, \mathbf{r}_2, \dots, \mathbf{r}_N) = V^{(0)} + \sum V^{(1)}(\mathbf{r}_i) + \sum_{i < j} V^{(2)}(\mathbf{r}_i, \mathbf{r}_j) + \sum_{i < j < k} V^{(3)}(\mathbf{r}_i, \mathbf{r}_j, \mathbf{r}_k) + \dots$$

- EAM expansion

$$V(\mathbf{r}_1, \mathbf{r}_2, \dots, \mathbf{r}_N) = \sum_i E(\rho_i) \quad \rho_i = \sum_n^{\text{neighbours}} \rho(r_{in})$$

Generating potentials

- How is an interatomic potential generated?
 - empirical, analytic formula
 - choose target properties (even forces)
 - fit free parameters to reproduce properties
 - hope that the formula remains reasonably valid everywhere in the configurational space
- The GAP way:
 - no fixed formula
 - search in the space of smooth functions
 - identify target configurations
 - fit to arbitrary precision QM data
 - extend target set if needed



GAP

- energy is sum of atomic energies
- $\epsilon(\mathbf{r}_1, \mathbf{r}_2, \mathbf{r}_3 \dots)$ is not practical
- matrix $\mathbf{r}_i \cdot \mathbf{r}_j$ is complete but not invariant to permutations
- symmetric polynomials are also complete but not invariant to rotation
 - CH_5^+ : all terms, chosen the rotationally invariant ones
- our solution
 - atomic energy is a functional of atomic density
 - express atomic density in rotationally invariant terms

GAP: function fitting

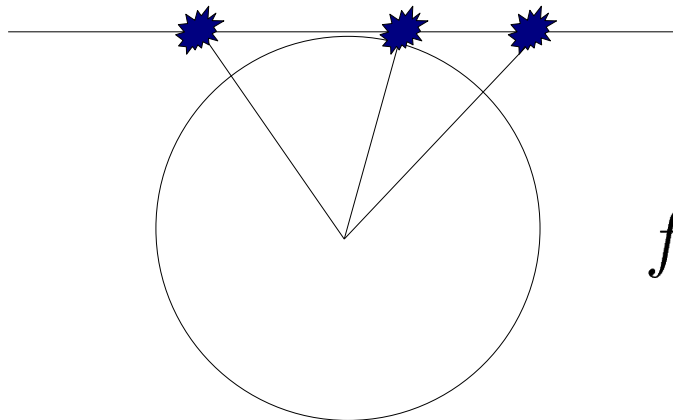
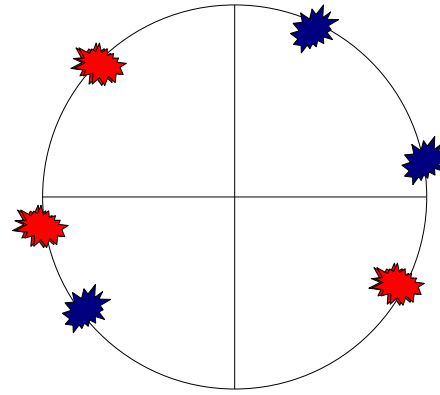
- how to choose the target configurations?
- an optimal way to interpolate many-dimensional functions

$$f(x) = \sum_i w_i \exp\left(-\frac{(x - x_i)^2}{2\sigma^2}\right)$$

- magic trick: finding which fitting points are optimal for reproducing a very large data set
- calculate accurate forces and energies (DFT)
- perform fit using sum of derivatives of ϵ_i (forces) and sum of ϵ_i (energy) as the target function

Invariant representations

- Two examples:
 - 1D periodic function
 - 2D object
- 1D: $f(\phi) \rightarrow \hat{f}_n$
- 2D: project on the Riemann-sphere then express it in spherical harmonics basis:



$$f(x, y) = f(\theta, \phi) \rightarrow f_{lm}$$

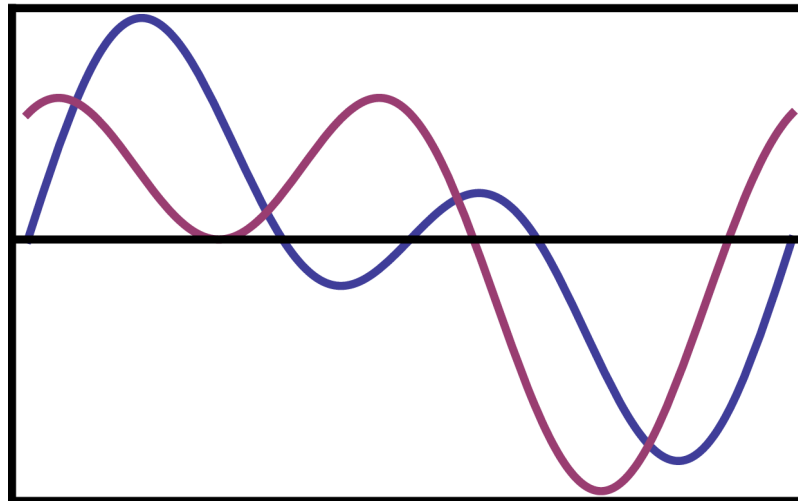
Invariant representations

- Power spectrum:

- 1D: $g(\phi) = f(\phi + \phi_0)$ $\hat{g}_n = \exp(in\phi_0)\hat{f}_n$
 $p_n = \hat{f}_n^* \hat{f}_n \rightarrow \hat{f}_n^* \exp(-in\phi_0) \hat{f}_n \exp(in\phi_0) = \hat{f}_n^* \hat{f}_n$

- 2D: $\mathbf{f}_l \rightarrow \mathbf{D}_l \mathbf{f}_l$ $\mathbf{f}_l^\dagger \mathbf{f}_l \rightarrow \mathbf{f}_l^\dagger \mathbf{D}_l^\dagger \mathbf{D}_l \mathbf{f}_l = \mathbf{f}_l^\dagger \mathbf{f}_l$

- incomplete representation, phase information lost



Invariant representations

- Bispectrum (almost complete):

- 1D: $b_n = \hat{f}_{n+m}^* \hat{f}_n \hat{f}_m \rightarrow$

$$\rightarrow \hat{f}_{n+m}^* \exp(-i(n+m)\phi_0) \hat{f}_n \exp(in\phi_0) \hat{f}_m \exp(im\phi_0) = \hat{f}_{n+m}^* \hat{f}_n \hat{f}_m$$

- 2D: $\mathbf{f}_{l_1} \otimes \mathbf{f}_{l_2} \rightarrow (\mathbf{D}_{l_1} \otimes \mathbf{D}_{l_2}) \mathbf{f}_{l_1} \otimes \mathbf{f}_{l_2}$

$$\mathbf{D}_{l_1} \otimes \mathbf{D}_{l_2} = \mathbf{C}_{l_1, l_2}^\dagger \left[\begin{array}{c} l_1 + l_2 \\ \bigoplus \\ l = |l_1 - l_2| \end{array} \mathbf{D}_l \right] \mathbf{C}_{l_1, l_2}$$

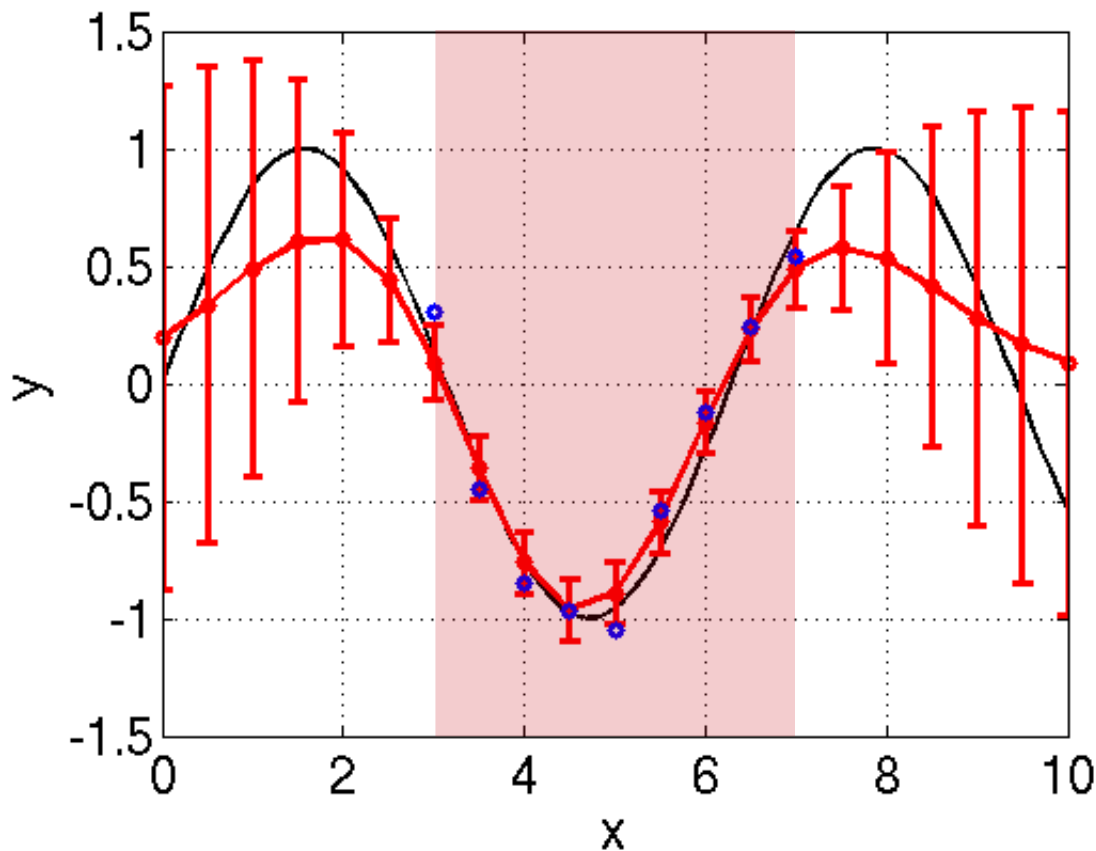
$$\mathbf{C}_{l_1, l_2} \mathbf{f}_{l_1} \otimes \mathbf{f}_{l_2} \rightarrow \left[\begin{array}{c} l_1 + l_2 \\ \bigoplus \\ l = |l_1 - l_2| \end{array} \mathbf{D}_l \right] \mathbf{C}_{l_1, l_2} \mathbf{f}_{l_1} \otimes \mathbf{f}_{l_2} = \bigoplus_{l = |l_1 - l_2|}^{l_1 + l_2} \mathbf{g}_{l_1, l_2, l}$$

$$p_{l_1, l_2, l} = \mathbf{f}_l^\dagger \mathbf{g}_{l_1, l_2, l}$$

Invariant representations: 3D objects

- project on a 4D sphere
- $f(x, y, z) \rightarrow f(\alpha, \theta, \phi)$
- θ and ϕ are the same as the 3D polar coordinates
- express in 4D spherical harmonics basis: the Wigner D-matrices
- bispectrum is analogous to 3D case
- 4D CG-coefficients are direct products of 3D CG-coefficients

GP: a simple interpolation



target function: $\sin x$

fit points between 3 and 7

random noise on fit points

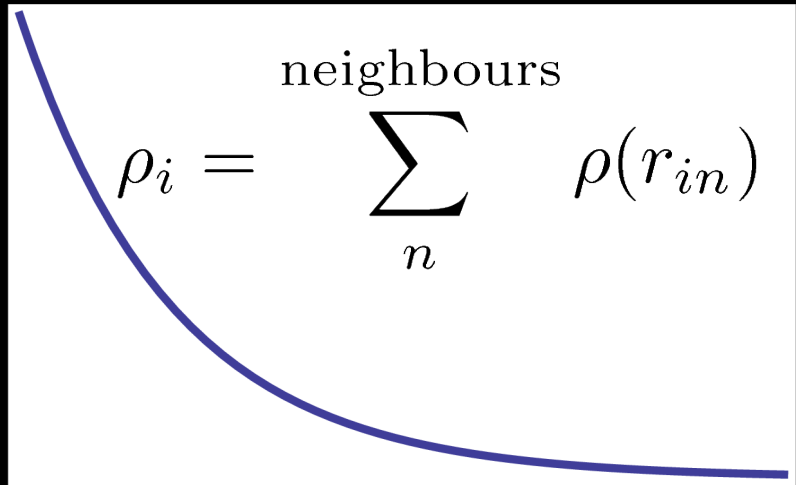
expectation value and variance predicted

Target configurations

- Hot MD of interesting systems
 - surfaces
 - interstitial
 - vacancy
 - quenched liquid
- energies and forces of configuration samples with DFT

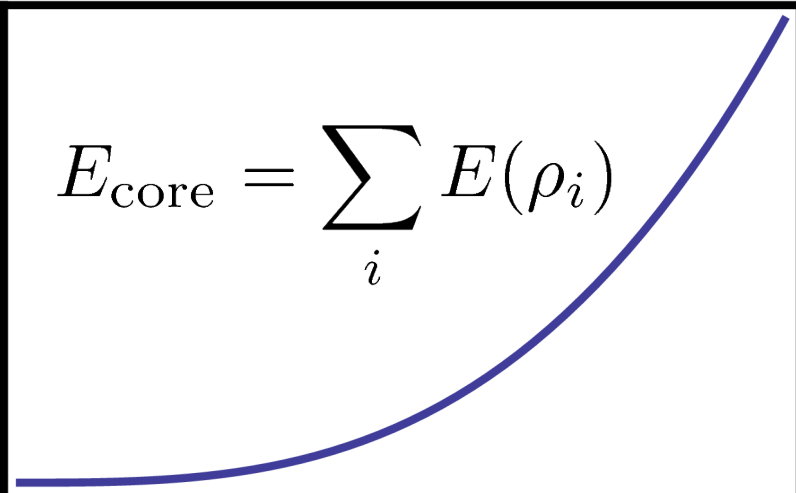
'Baseline' potential

- GP gives 0 as an answer when unsure
- add core repulsion to correct for very close atoms
 - EAM formula
 - fitted to high pressure DFT results
- add dispersion term for long-range interactions



A graph showing a blue curve that starts at a high value on the y-axis and decays rapidly towards the x-axis, representing a potential energy function that decreases as distance increases.

$$\rho_i = \sum_n^{\text{neighbours}} \rho(r_{in})$$



A graph showing a blue curve that starts near zero on the y-axis and increases exponentially as it moves along the x-axis, representing a potential energy function that increases as distance decreases.

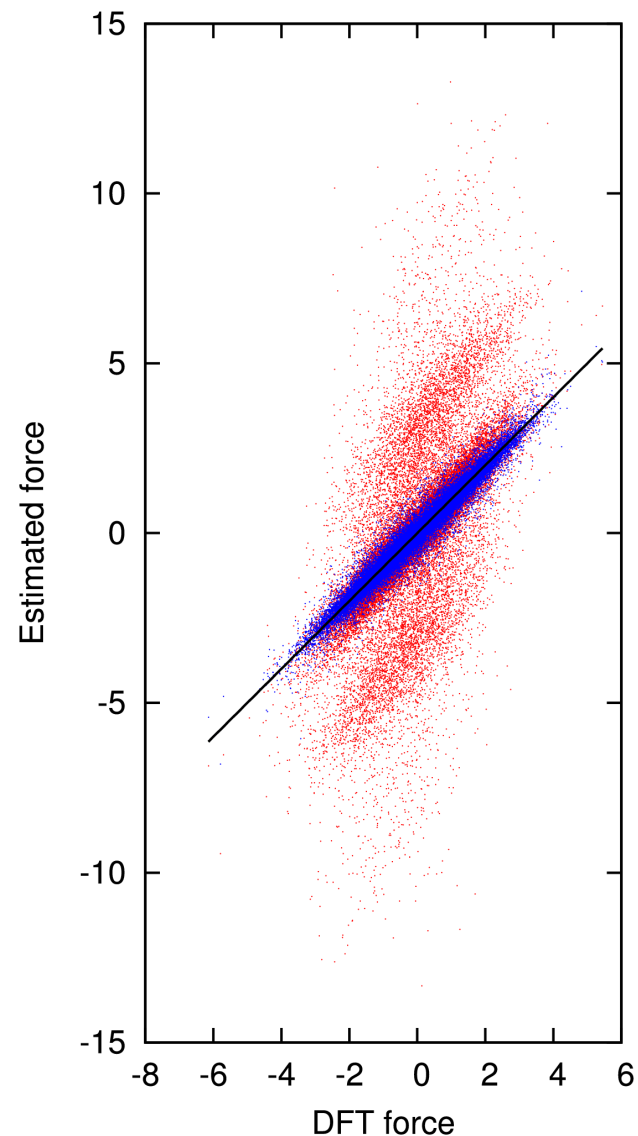
$$E_{\text{core}} = \sum_i E(\rho_i)$$

GAP for carbon

- target configurations
 - bulk phases
 - transition from diamond to graphite
 - 111 and 100 surfaces
 - vacancy
 - interstitial
 - amorphous carbon

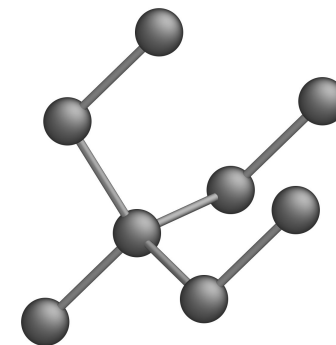
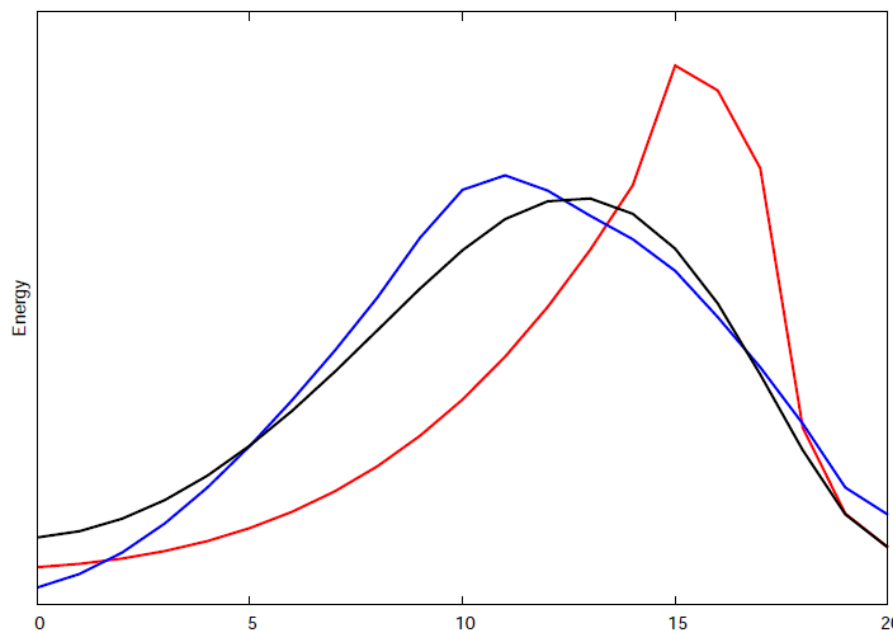
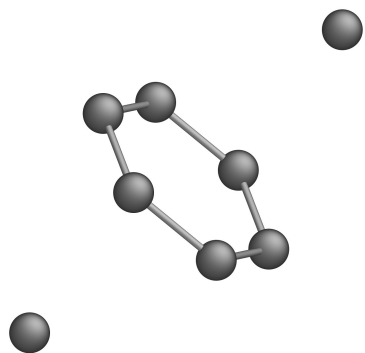
GAP test

- force correlation
DFT vs (REBO, GAP)
- $\sigma = 0.26$ eV/Å
- 100 teaching points
- 2.5 Å cutoff



GAP test

- diamond to graphite transition along arbitrary reaction coordinate
- energy from DFT, **REBO** and **GAP**



Defining local energies

- how to obtain local energy from DFT
 - surface energy
 - visualisation
- best fit potential: use local energy
- equivalent to chemical potential of an atom
 - replacing atoms to infinity

Approximate local energies

- restricted part of configurational space
 - use force and energy information only along a minimisation from 'gas' phase or
 - blow up a configuration
- use it as post-processing tool
- visualise 'hot' atoms

Summary

- interpolation in high-dimensional space via GP
- target is quantum mechanical data
- extendable method
- carbon potential that captures sp^2 - sp^3 transition
- new approach in defining local energies

- more than one atom type: $\rho(\mathbf{r}) = \sum_i c_i \delta(\mathbf{r} - \mathbf{r}_i)$
- electrostatics: subtract Coulomb energy