

ELECTRONIC STRUCTURE METHODOLOGY 2

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Lecture Three

THE KOHN-SHAM EQUATIONS

$$\begin{aligned} E_{\text{tot}}[\{\psi_i\}] = & -2 \sum_i \frac{\hbar^2}{2m} \int \psi_i \nabla^2 \psi_i d\mathbf{r} \\ & + \int V_{\text{ion}}(\mathbf{r}) n(\mathbf{r}) d\mathbf{r} \\ & + \frac{e^2}{2} \int \frac{n(\mathbf{r}) n(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} d\mathbf{r} d\mathbf{r}' \\ & + E_{\text{xc}}[n(\mathbf{r})] + E_{\text{ion}}(\mathbf{R}_i) \end{aligned}$$

The charge density is written as:

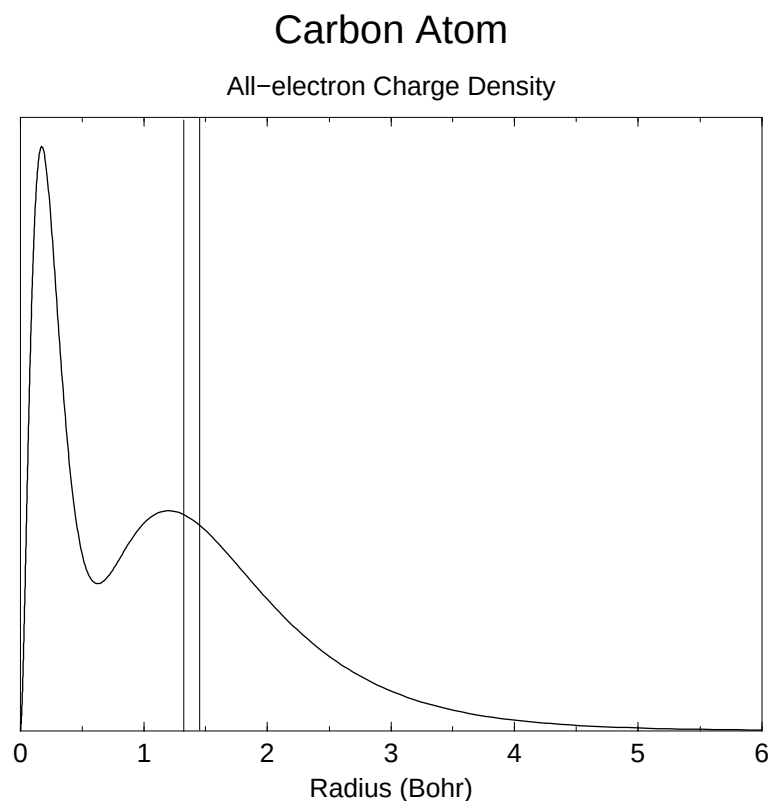
$$n(\mathbf{r}) = 2 \sum_i |\psi_i(\mathbf{r})|^2$$

The potentials are given by *functional* derivatives:

$$V_{\text{xc}}(\mathbf{r}) = \frac{\delta E_{\text{xc}}[n(\mathbf{r})]}{\delta n(\mathbf{r})} \text{ etc.}$$

$$\left[-\frac{\hbar^2}{2m} \nabla^2 + V_{\text{ion}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r}) + V_{\text{xc}}(\mathbf{r}) \right] \psi_i(\mathbf{r}) = \epsilon_i \psi_i(\mathbf{r})$$

CORE AND VALENCE ELECTRONS

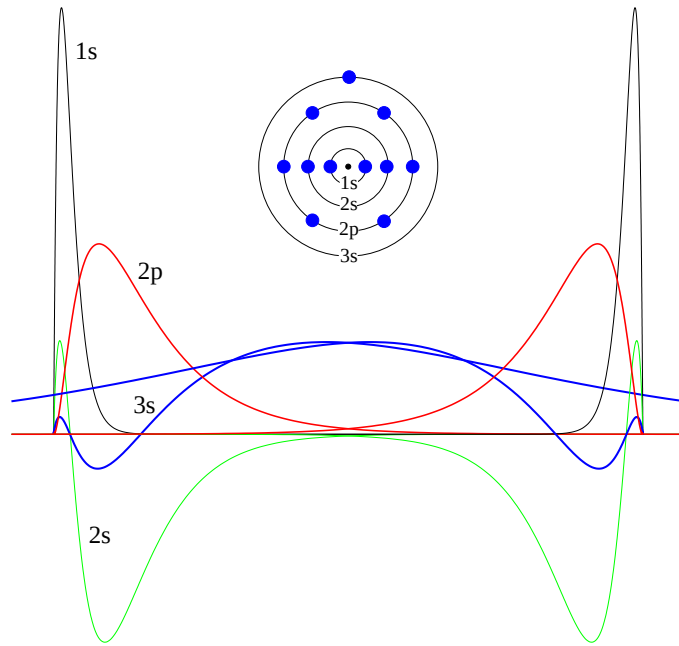


- Core electrons don't take part in bonding (definition!)

Level	Energy(Ry)	Occupation
1s	-19.90408	2.000
2s	-1.00279	2.000
2p	-0.39838	2.000

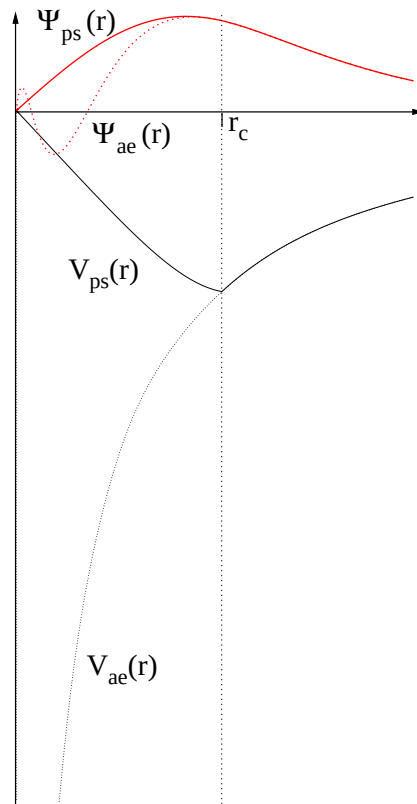
- To avoid calculating the properties of the (possibly many) core electrons we can invent *pseudopotentials*

THE PSEUDOPOTENTIAL APPROACH



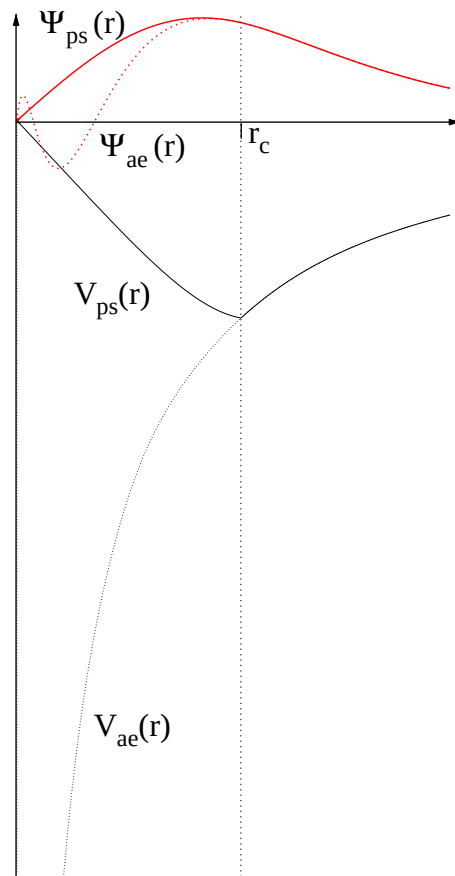
- The pseudopotential was originally thought of as the weaker effective potential due to the orthogonality of the valence to the core wavefunctions
- This concept provided a justification and theoretical framework for the nearly free electron approximation
- The true power of the method unleashed by much later *ab initio* formulations

THE PSEUDOPOTENTIAL SCHEME



- Do an all electron atomic calculation
- Choose a core radius, and *pseudise* the atomic wavefunctions in the channels of interest
- The simplest schemes conserve the *norm* but this can be relaxed
- Invert the Schrödinger equation to find the *pseudopotential*

THE PSEUDOPOTENTIAL SCHEME



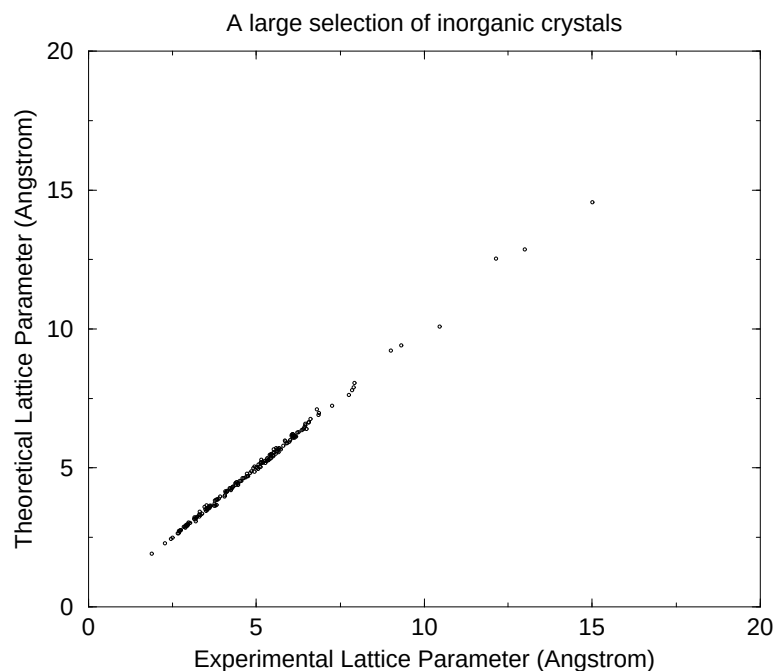
- The resulting pseudopotential will reproduce the *reference* eigenstates by construction, and is nonlocal in general
- *Transferability* is of great importance – the potentials should be tested extensively
- There may be *ghost states*

IT WORKS!

Material	Expt	Theory	Delta	Type
LaBi	6.57	6.648	1.2%	alloy
CaF ₂	5.4626	5.496	0.6%	halide
Ag	4.086	4.112	0.6%	metal
V	3.028	3.019	-0.3%	metal
ZrN	4.62	4.634	0.3%	misc
NbO	4.2103	4.2344	0.6%	oxide
GaAs	5.653	5.663	0.2%	semiconductor
CoSi ₂	5.36	5.3	-1.1%	silicide

A REMINDER

A comparison of theory with experiment



- We got all this from
 - Schrödinger's Equation
 - Density functional theory
 - a many-body uniform electron gas
 - some clever approximations

Milman, Winkler, White, Pickard, Payne, Akhmatkaya, and Nobes.

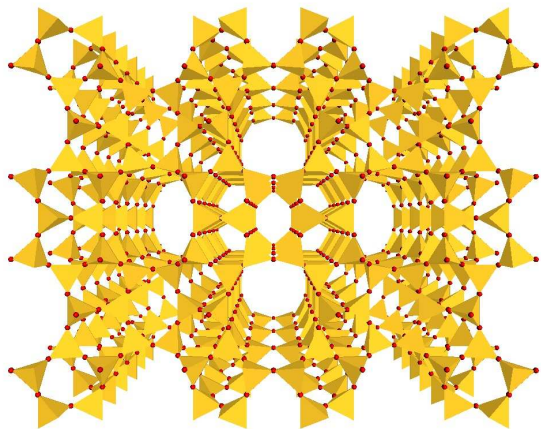
Electronic structure, properties and phase stability of inorganic crystals:

The pseudopotential plane-wave approach.

International Journal of Quantum Chemistry, **77**:895-910, 2000.

WE MADE LOTS OF PROGRESS BUT . . .

. . . these approximations are **not enough** to allow calculations for large systems



Ferrierite

- Conventional matrix diagonalisation
 - too slow for a large, well converged, basis set
 - calculates too many (unoccupied) states
- The unoccupied states do not contribute to the total energy, so why calculate them?

CALCULATING THE TOTAL ENERGY

1. Eigenvalue term
2. Potential term
3. Ion-ion term

To get eigenvalues we use:

$$H|\psi_i\rangle = \epsilon_i|\psi_i\rangle$$

$$H = -\frac{1}{2}\nabla^2 + V(\mathbf{r})$$

- Most stable computational approach is to **directly minimise** the total energy
- **Do not** solve $H|\psi_i\rangle = \epsilon_i|\psi_i\rangle$ directly
- Use iterative diagonalisation for just the states that we need
- We still need to operate H on $|\psi_i\rangle$

OPERATING WITH H ON $|\psi_i\rangle$

- H divides into two parts
 - The potential — diagonal in **real** space
 - The kinetic energy — diagonal in **reciprocal** space
- The potential $V(\mathbf{r})$ can be evaluated in **real** space
- Kinetic energy in **reciprocal** space — $\frac{1}{2}|\mathbf{k} + \mathbf{G}|^2$ if we use a plane wave basis set (more on this later)
- Could use FFTs (Fast Fourier Transforms) and evaluate each term in appropriate space
- Different basis sets suggest different strategies

EVALUATING THE ENERGY

- The eigenvalue sum:

$$\epsilon_i = \langle \psi_i | H | \psi_i \rangle$$

- Potential energy — product of potential with density, then integrate :

$$E_{\text{pot}} = \int V(\mathbf{r})n(\mathbf{r})d\mathbf{r}$$

- $E_{\text{ion-ion}}$ converges slowly in both real and reciprocal space so use the Ewald identity
 - splits the sum between two spaces
 - the sum of the terms converges rapidly

ITERATIVE DIAGONALISATION

- Need energy **gradient** for each band i and iteration m :
- Enforce orthogonality via the gradient:

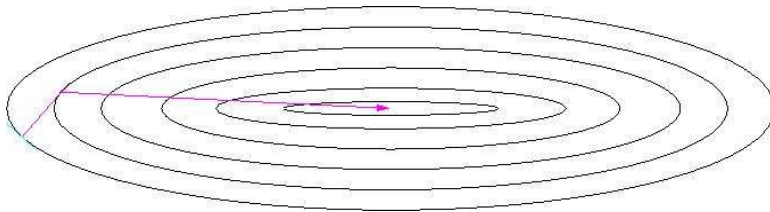
$$|\eta_i^m\rangle = -(H - \epsilon_i^m)|\psi_i^m\rangle$$

$$|\eta_i'^m\rangle = |\eta_i^m\rangle - \sum_j \langle\psi_j|\eta_i^m\rangle|\psi_j\rangle$$

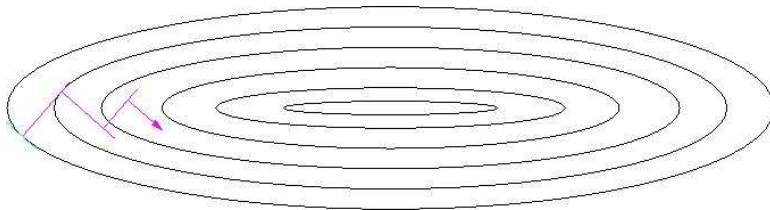
$$\epsilon_i^m = \langle\psi_i^m|H|\psi_i^m\rangle$$

- **Antisymmetry** of wavefunctions $\rightarrow$ **orthogonality** of bands at each k-point
- Orthogonalisation is costly and **dominates** in the limit of a very large system

MINIMISATION



- Steepest descents — safe, but very slow



- Conjugate gradients — use history to ensure independence (exact for quadratic functions)
- Preconditioning — to encourage all components of the wavefunction to converge at comparable rates