

CASTEP Workshop 2006

Finding the Kohn-Sham
Groundstate

University of York

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Overview

- The DFT Schrödinger equation
- Minimising the KS energy
- Metals and finite temperatures
- Density mixing and Ensemble DFT

The DFT Hamiltonian

- The object is to find the single-particle solutions to the Schrödinger equation

$$\hat{H}_{ks} \Psi_i = \epsilon_i \Psi_i$$

where i labels the different eigenstates and the Hamiltonian is given by

$$\hat{H}_{ks} = -\frac{1}{2} \nabla^2 + \hat{V}_H + \hat{V}_{xc} + \hat{V}_{ps}$$

- We are using a plane-wave basis and sampling the Brillouin zone at a finite number of **k-points**,

$$\Psi = \sum_{\mathbf{G}} c_{\mathbf{k}, \mathbf{G}} e^{i(\mathbf{k} + \mathbf{G}) \cdot \mathbf{r}}$$

where the sum is over all reciprocal lattice vectors below some **cut-off** wavevector.

- The wavefunctions at different **k-points** only interact via the density, so each **k-point** yields a separate Kohn-Sham equation.

- For simplicity we will consider only a calculation at a single \mathbf{k} -point, and omit the subscript \mathbf{k} .
- Each eigenstate is normalised to contain only one particle.

$$\langle \Psi_i | \Psi_i \rangle = 1$$

- Each eigenstate is orthogonal to all other eigenstates (Pauli exclusion principle).

$$\langle \Psi_i | \Psi_j \rangle = 0$$

Self-Consistency

- The Hartree potential represents the Coulomb repulsion between the different particles and is given by:

$$V_H(r) = \int \frac{n(r')}{|r-r'|} d^3 r'$$

where n is the charge density

$$n(r) = \sum_i f_i |\Psi_i(r)|^2$$

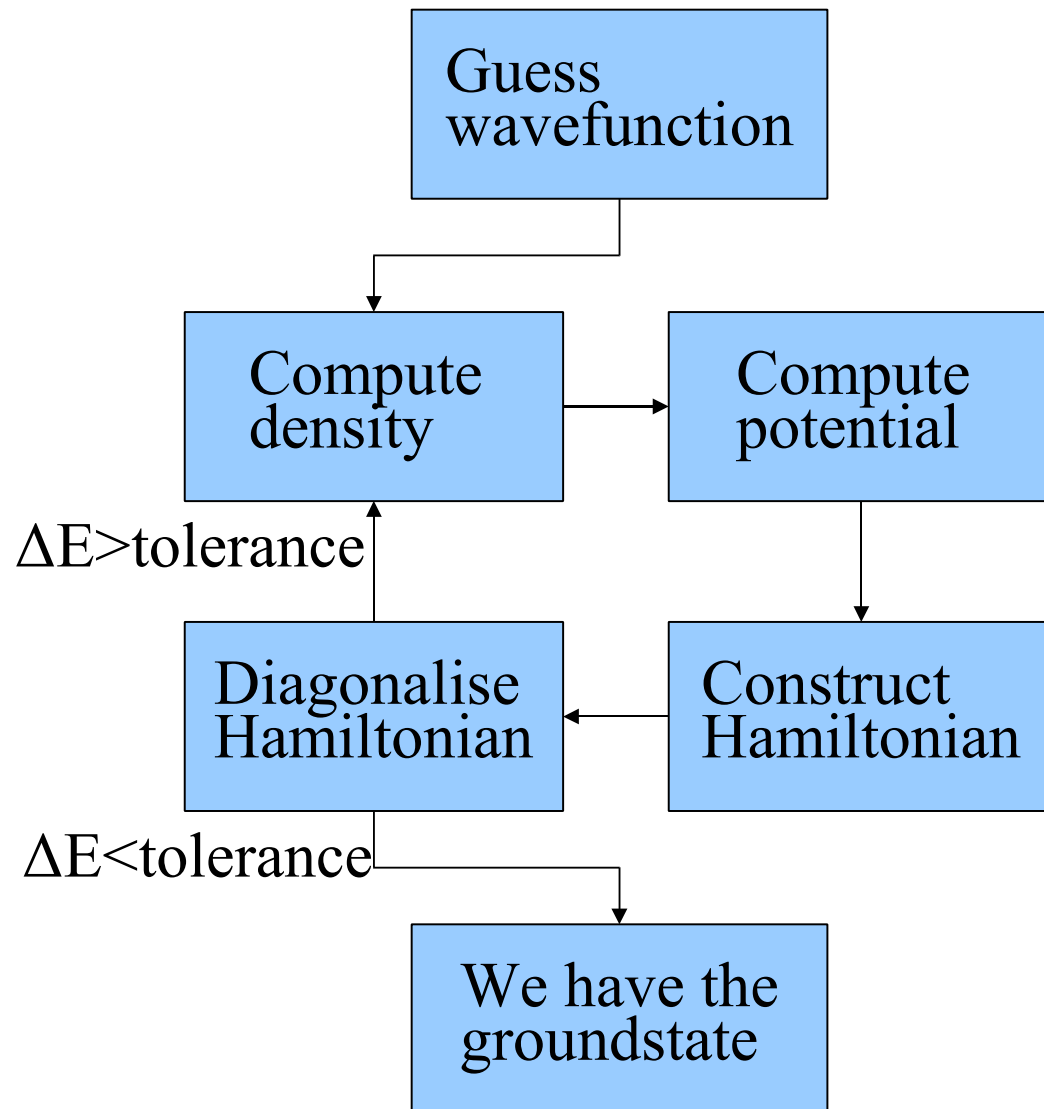
- Notice that the Hartree potential depends on the density, and the density depends on the wavefunctions, we no longer have a fixed Hamiltonian to solve.
- The exchange-correlation potential also depends on the density – the Hamiltonian changes as the wavefunctions change!
- We need to find a set of wavefunctions that lead to a Hamiltonian whose solutions are the wavefunctions we started with.
- This is called **self-consistency**.

Putting it on a computer

- The wavefunction is represented as a vector formed from the complex coefficients c .
- Operators such as the Hamiltonian are linear transformations of wavefunctions, and are represented by square Hermitian matrices.
- In principle we can now solve the Kohn-Sham equations by constructing and then diagonalising this Hamiltonian matrix to get all of the eigenstates and eigenvalues.

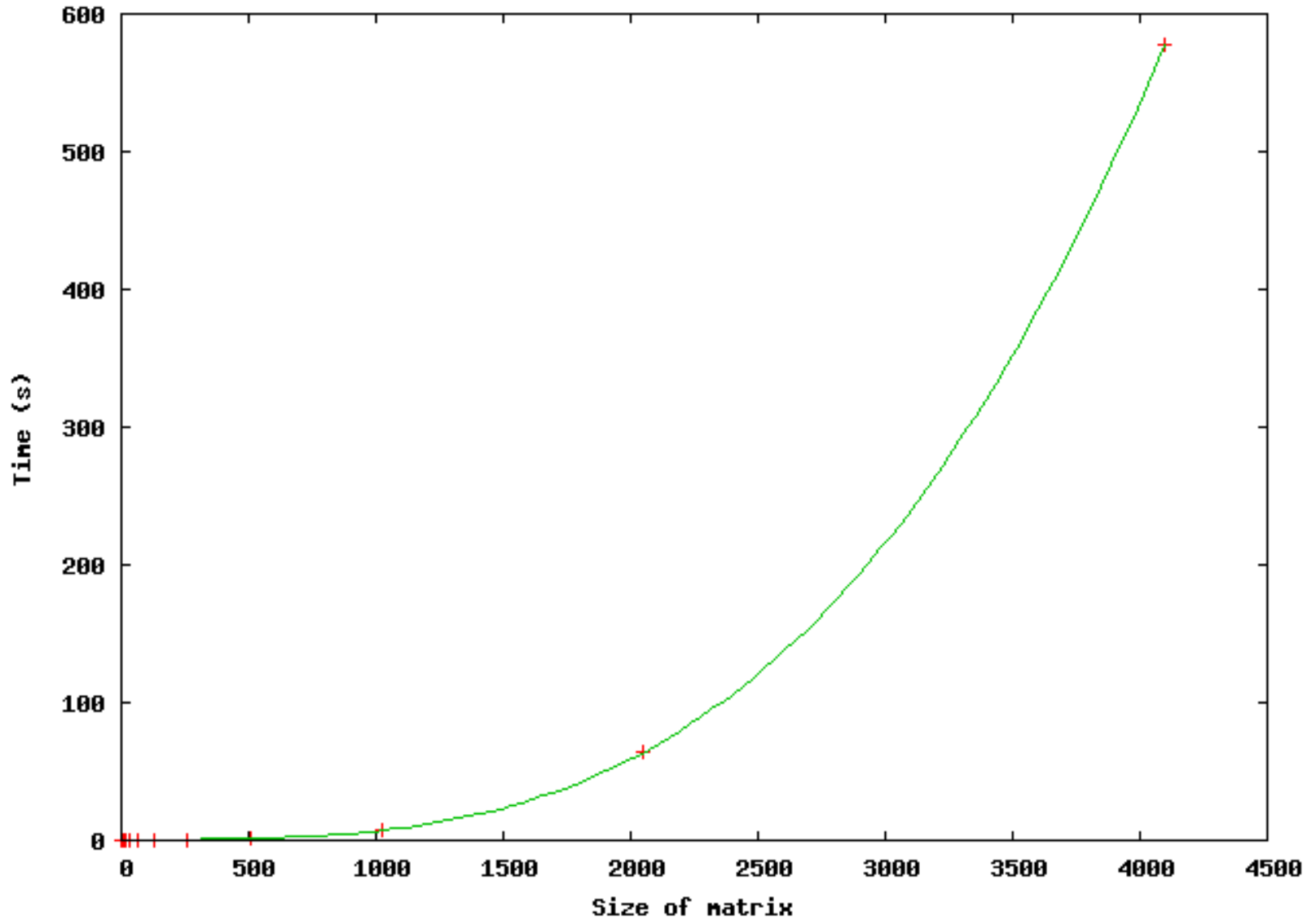
First Attempt - Diagonalisation

- Start with an initial set of wavefunctions, and construct the density.
- Construct the local potential from the density, and hence the Hamiltonian.
- Diagonalise to get eigenstates.
- Construct new density from eigenstates, and thus a new Hamiltonian.
- Repeat until the new Hamiltonian is the same as the old one – **self-consistent field (SCF)**.



- Unfortunately the number of plane-waves required is large, often 10,000-100,000.
- Suppose we require 30,000 plane-waves for a calculation. The Hamiltonian is a matrix of size $30,000 \times 30,000 = 9 \times 10^8$ elements.
- The computational time required to diagonalise a matrix scales as the cube of its size...

Computation Time for Direct Diagonalisation



- Estimated time to diagonalise our Hamiltonian on a 2.8GHz Pentium IV is 74 hours (per \mathbf{k} -point, per SCF cycle).
- This procedure gives us all 30,000 eigenstates, but typically fewer than 1% of these are occupied; the vast majority do not contribute to the density and have little physical meaning.

Minimising the KS energy

- An alternative method is to improve the wavefunctions iteratively. The energy of a wavefunction can be calculated as:

$$E = \sum_i \langle \Psi_i | \hat{H}_{ks} | \Psi_i \rangle$$

- We can differentiate this to find the gradient of the energy w.r.t. the wavefunction.

$$\frac{\delta E}{\delta \langle \Psi_i |} = \hat{H}_{ks} | \Psi_i \rangle$$

Steepest descent

- Using Lagrange multipliers we can include the constraint of normalisation and show that the constrained gradient is

$$\hat{G}|\Psi_i\rangle = \hat{H}|\Psi_i\rangle - \epsilon_i|\Psi_i\rangle$$

- If we include the orthogonalisation constraint as well we get:

$$\hat{G}|\Psi_i\rangle = \hat{H}|\Psi_i\rangle - \sum_j \langle \Psi_j | \hat{H} | \Psi_i \rangle |\Psi_j\rangle$$

- We can now take any wavefunction and calculate the energy gradient direction.
- This is the direction in which the energy *increases most rapidly*. Since we are looking for the lowest energy we search in the opposite direction, the *direction of steepest descent*.
- The *constrained gradient* is used rather than the ordinary gradient to prevent all the eigenstates converging to the lowest one—only one particle is allowed in each eigenstate.

The Hamiltonian

- In principle the Hamiltonian is still of size number of plane-waves squared, so difficult to store in RAM and expensive to apply.
- Fortunately the kinetic energy operator is diagonal in reciprocal space, and the local (Hartree+exchange-correlation) potential is diagonal in real-space, so we only actually need to store these diagonal components.

- Now start with initial wavefunction and construct the steepest descent direction.

$$|\Phi\rangle = -\hat{G}|\Psi\rangle$$

- Search along this direction for lowest energy.

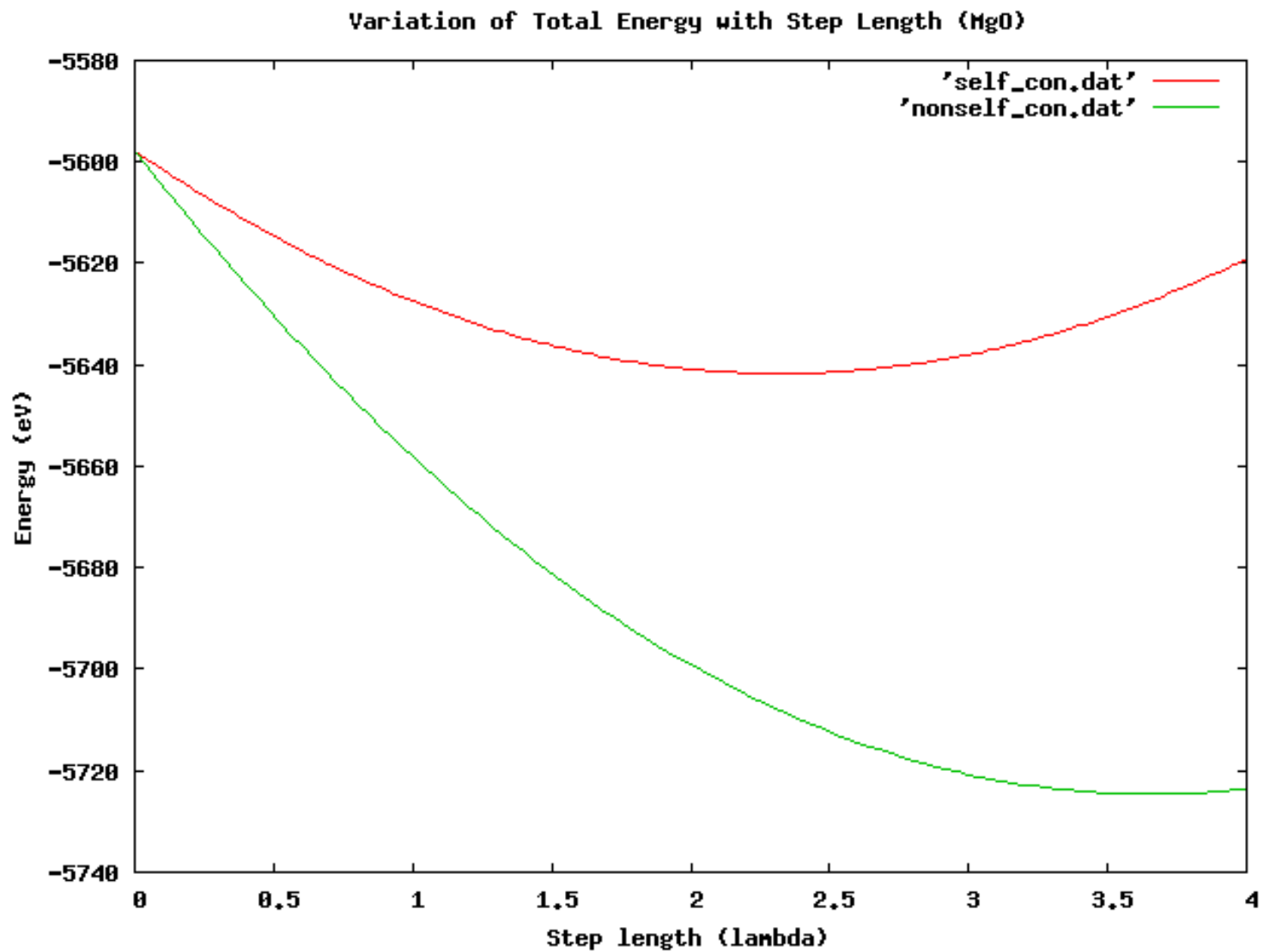
$$|\Psi'\rangle = |\Psi\rangle + \lambda|\Phi\rangle$$

- Calculate new search direction and repeat. After the first step better schemes may be used, such as **conjugate gradients**.

- Iterative diagonalisation is a bit slower per eigenstate than direct diagonalisation, but allows us to calculate only the $\sim 1\%$ of eigenstates that actually have particles in them.
- For N electrons we only need N eigenstates, so we iteratively diagonalise the Hamiltonian to obtain the N lowest eigenstates.
- E.g. 8-atom magnesium oxide cell at 330eV has 1021 plane-waves and 56 electrons. Direct diagonalisation time is 7.93s per \mathbf{k} -point per SCF, iterative diagonalisation is 0.48s.

Self-consistency revisited

- We're now updating the wavefunctions iteratively. At what point do we recalculate the density?
- **At each point along λ ?** This is called a self-consistent method. In general these are slow, but stable.
- **After a certain number of wavefunction updates?** This is called a non-self-consistent method. In general these are fast, but unstable.



- The non-self-consistent minimum is usually found at larger step sizes than the self-consistent one, so non-self-consistent methods often overshoot the minimum.
- Consistent overshooting causes a phenomenon known as **charge sloshing** (see later).
- Charge sloshing is worse for large cells — consider the Hartree potential again but this time in reciprocal space:

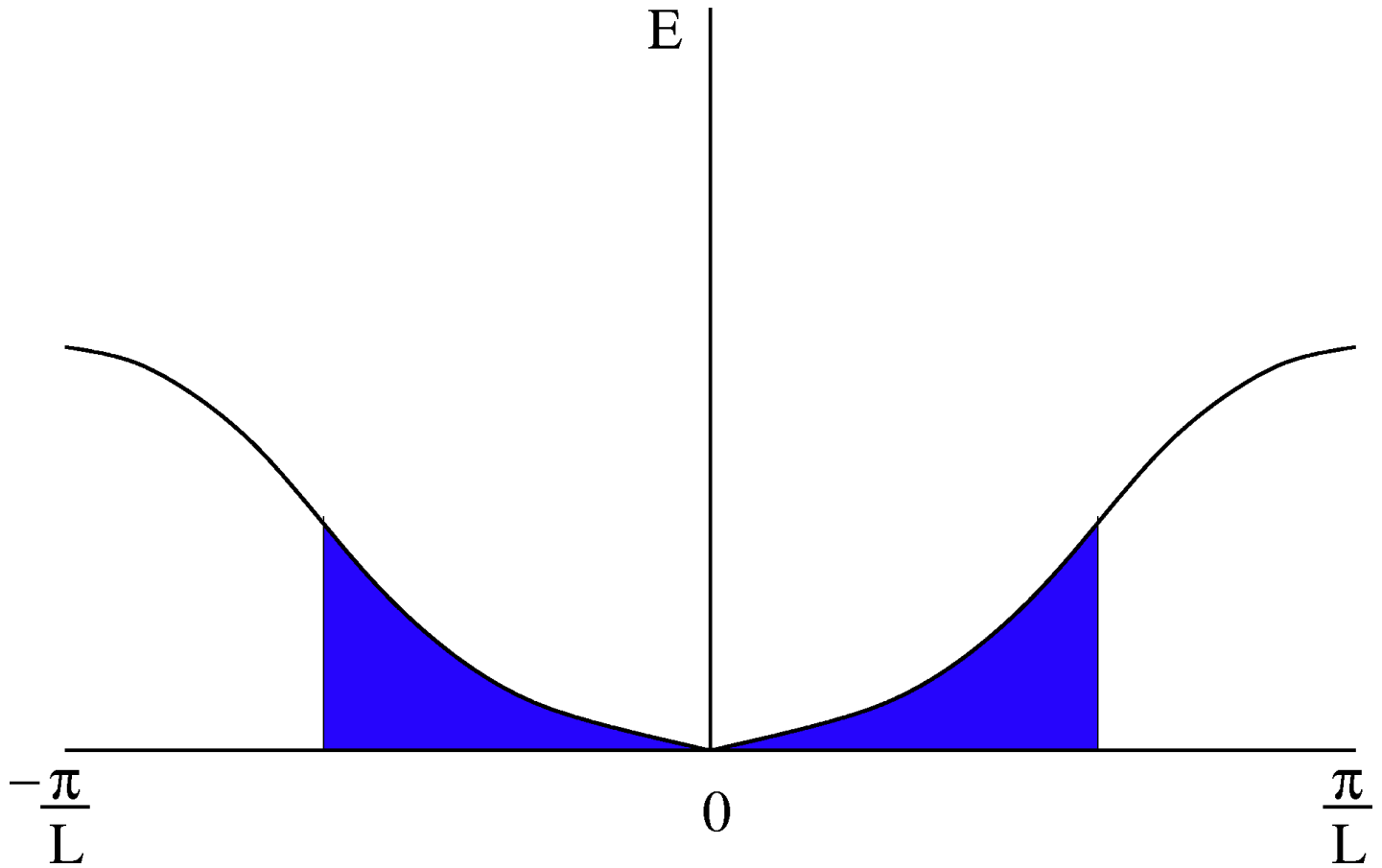
$$V_H(\mathbf{G}) = \frac{4\pi n(\mathbf{G})}{\Omega_{cell} |\mathbf{G}|^2}$$

Large cell means small G-vectors

Metallic systems

- When we construct the density, we do so from the lowest N eigenstates.
- As the band-gap decreases, it becomes increasingly difficult to identify the lowest eigenstates (i.e. to find the Fermi surface).
- Bands below the Fermi energy are completely occupied and contribute fully to the density; bands above are empty and do not contribute at all.

k-points and metals



Sloshing

- Suppose band A is below the Fermi surface and B above. It is possible that by occupying band A we change the density such that bands A and B swap in energy.
- We now occupy band B instead and empty band A, whereupon the density changes such that the original ordering is restored.
- We will never converge to the groundstate!
This is known as a **sloshing instability**.

The Mermin functional

- Sloshing arises from the discontinuous occupation of the eigenstates across the Fermi surface at zero temperature.
- Mermin smeared the Fermi surface, exactly like heating the system up.
- Eigenstates near the Fermi surface are now neither full nor empty, but partially filled.
- There is an entropy contribution to the energy from the partial occupancies.

Occupancies

- We assign the occupancies based on the eigenvalues and the Fermi energy.
- The density depends on the occupancies as well as the wavefunctions, so changing the occupancies changes the density, and hence the Hamiltonian.
- Changing the Hamiltonian changes the eigenvalues of the states, so yet again we need iterate to **self-consistency**.

Ensemble DFT (EDFT)

- In EDFT the density is reconstructed whenever the wavefunctions or occupancies change, and both the wavefunctions and occupancies are only changed when it would lower the total energy.
- Each SCF cycle consists of a number of wavefunction updates (usually five), and after each wavefunction update the occupancies are also updated.

Ensemble DFT (EDFT)

- Constructing the density requires a Fourier transform of the wavefunctions into real-space, and this can be slow (especially in parallel).
- EDFT is:
 - Variational (energy decreases at each step)
 - Stable
 - Slow

EDFT parameters

- EDFT is controlled by the .param file
 - `elec_method: EDFT`
- Several occupancy cycles are performed every step to find the optimum partial occupancies.
 - `num_occ_cycles: 6`

for some systems you can reduce the number of occupancy cycles to speed up the calculation. 3 or 4 is often fine, even 2 is OK for simple systems.

Density mixing (DM)

- In DM the density is only reconstructed once per SCF cycle. To prevent sloshing instabilities this new density is mixed with a fraction of the previous densities.
- A typical DM SCF cycle will update each trial eigenstate two or three times, reassign occupancies, and only then construct the density.

Density mixing (DM)

- DM is:
 - Non-variational
 - Potentially unstable
 - Quick
- The key to the stability of DM is the manner of the mixing. The best schemes in Castep are the Pulay and Broyden schemes.

DM parameters

- ❑ DM is controlled by the .param file
 - `elec_method: DM`
- ❑ The number of past densities to store
 - `mix_history_length: 7`
- ❑ The DM scheme
 - `mixing_scheme: Pulay` or `Broyden`
- ❑ The initial mixing amplitudes
 - `mix_charge_amp: 0.7` (0.5 in Materials Studio)
 - `mix_spin_amp: 2.0`

decrease for a slight increase in stability.

DM parameters

- The screening wavevectors—the mixing schemes work by building up an approximation to the system's dielectric. The initial approximation is of the form

$$K(q) = \frac{q^2}{q^2 + q_c^2}$$

- `mix_charge_gmax`: 1.5 1/ang
- `mix_spin_gmax`: 1.5 1/ang

Large frequencies

- Recall that the sloshing instabilities were due to the low frequency (small q) components. What about the high-frequency components?
- Usually we accept the high frequency part of the new density without mixing it at all
- At what point is a frequency “high enough” to not mix? It’s controlled by the parameter
 - `mix_cut_off_energy`
- This defaults to the wavefunction cut-off (`cut_off_energy`).

General parameters

- ❑ The number of wavefunction updates per SCF
 - `max_sd_steps`: 1
 - `max_cg_steps`: 4DM may not perform all of these steps.
- ❑ The energy to smear the Fermi level by
 - `smearing_width`: 0.2 eV
- ❑ The number of extra bands to include
 - `nextra_bands`Including extra bands slows the SCF cycles, but may improve convergence (i.e. take fewer SCFs).

EDFT output (iprint: 1)

SCF loop	Energy	Fermi energy	Energy gain per atom	Timer (sec)	<-- SCF
Initial	-5.60447982E+003	6.04960700E+000		5.57	<-- SCF
1	-5.65487212E+003	8.54480709E+000	6.29903671E+000	34.10	<-- SCF
2	-5.65488381E+003	8.55197524E+000	1.46193897E-003	62.10	<-- SCF
3	-5.65488514E+003	8.54531708E+000	1.65970615E-004	90.12	<-- SCF
4	-5.65488559E+003	8.54559484E+000	5.64376170E-005	120.43	<-- SCF
5	-5.65488567E+003	8.54576989E+000	9.21133649E-006	149.52	<-- SCF
6	-5.65488568E+003	8.54568880E+000	1.14975580E-006	179.38	<-- SCF
7	-5.65488568E+003	8.54574147E+000	1.12351628E-007	210.01	<-- SCF
8	-5.65488568E+003	8.54573805E+000	3.75314590E-008	240.19	<-- SCF
9	-5.65488568E+003	8.54573963E+000	1.34522225E-008	270.22	<-- SCF
10	-5.65488568E+003	8.54573949E+000	4.91008537E-009	300.98	<-- SCF
11	-5.65488568E+003	8.54573975E+000	1.66028364E-009	330.80	<-- SCF
12	-5.65488568E+003	8.54573998E+000	7.05528707E-010	358.88	<-- SCF
13	-5.65488568E+003	8.54573998E+000	1.69276619E-010	387.17	<-- SCF
14	-5.65488568E+003	8.54574005E+000	5.12373547E-011	416.85	<-- SCF

EDFT output (iprint: 2)

```
+----- WAVEFUNCTION LINE MINIMISATION -----+<- line
| Initial energy = -5.604480E+003 eV; initial dE/dstep = -3.982E+001 eV |<- line
|                               |<- line
|           1st step      2nd step      3rd step      4th step      5th step      |<- line
+-----+-----+-----+-----+-----+-----+<- line
| step | 1.500E+00  2.352E+00* unnecessary unnecessary unnecessary |<- line
| gain | 4.069E+01  4.658E+01* unnecessary unnecessary unnecessary |<- line
+-----+-----+-----+-----+-----+-----+<- line
* indicates the final, accepted state (should have the lowest energy)
```

```
+----- OCCUPANCY LINE MINIMISATION -----+<- EDFT
| Initial energy for occupancy cycle 1 = -5.651E+003 eV |<- EDFT
|                               |<- EDFT
|           1st step      2nd step      3rd step      4th step      5th step      |<- EDFT
+-----+-----+-----+-----+-----+-----+<- EDFT
| step | 1.000E+00  6.200E-01  8.783E-01  8.656E-01* unnecessary |<- EDFT
| gain | 1.940E+00  1.807E+00  2.007E+00  2.007E+00* unnecessary |<- EDFT
+-----+-----+-----+-----+-----+-----+<- EDFT
* indicates the final, accepted state (should have the lowest energy)
Final energy = -5.653E+003eV (change of 2.007E+000eV) for occ. cycle 1 <- EDFT
```

DM output (iprint: 1)

SCF loop	Energy	Fermi energy	Energy gain per atom	Timer (sec)	<-- SCF
Initial	-5.72401759E+003	7.33403352E+000		4.93	<-- SCF
1	-5.66310821E+003	7.15206775E+000	-7.61367268E+000	5.44	<-- SCF
2	-5.66315105E+003	7.15066794E+000	5.35423262E-003	6.16	<-- SCF
3	-5.65904032E+003	9.04183988E+000	-5.13841407E-001	7.43	<-- SCF
4	-5.65484948E+003	8.35292985E+000	-5.23854073E-001	8.71	<-- SCF
5	-5.65487927E+003	8.47113111E+000	3.72385144E-003	10.21	<-- SCF
6	-5.65488530E+003	8.53910153E+000	7.53186999E-004	11.66	<-- SCF
7	-5.65488567E+003	8.54432111E+000	4.59464638E-005	13.30	<-- SCF
8	-5.65488568E+003	8.54547827E+000	1.23017408E-006	14.61	<-- SCF
9	-5.65488568E+003	8.54569517E+000	4.93767620E-008	16.05	<-- SCF
10	-5.65488568E+003	8.54573154E+000	-1.00753908E-009	17.34	<-- SCF
11	-5.65488568E+003	8.54574500E+000	-6.34666478E-010	18.53	<-- SCF
12	-5.65488568E+003	8.54574128E+000	2.38785408E-011	19.61	<-- SCF
13	-5.65488568E+003	8.54574299E+000	8.01429567E-011	20.70	<-- SCF

DM output (iprint: 2)

Band	Steps	Eigenvalue		Change (energy drop)	stop cond	<- line
		Initial	Final			
1	2	-3.309434E+01	-3.322212E+01	1.2778E-01	--X	<- line
2	2	-3.303377E+01	-3.322107E+01	1.8730E-01	--X	<- line
3	2	-3.299567E+01	-3.321013E+01	2.1446E-01	--X	<- line
4	2	-3.292534E+01	-3.314784E+01	2.2249E-01	--X	<- line
5	2	-3.281925E+01	-3.314753E+01	3.2828E-01	--X	<- line
6	2	-3.273043E+01	-3.312386E+01	3.9343E-01	--X	<- line
7	2	-3.267706E+01	-3.308311E+01	4.0606E-01	--X	<- line
8	2	-3.259183E+01	-3.308239E+01	4.9056E-01	--X	<- line
9	2	-3.256198E+01	-3.306716E+01	5.0518E-01	--X	<- line
10	2	-3.253910E+01	-3.306556E+01	5.2646E-01	--X	<- line
11	2	-3.252031E+01	-3.306529E+01	5.4498E-01	--X	<- line

max no. steps performed <-----+|| <- line
elec eigenvalue tol reached <-----+| <- line
gradient flattened out <-----+ <- line

EDFT vs DM (Part I)

- Test system: 8-atom MgO (340eV)
- EDFT converges in 14 SCF cycles (416.9s)
 - 80 wavefunction updates
 - 328 density calculations
- DM converges in 13 SCF cycles (20.7s)
 - 50 wavefunction updates
 - 13 density calculations
- DM is the clear winner.

EDFT vs DM (Part I)

- Look at the initial energies; the Fermi energy in DM is much better than the EDFT one. Why?
- In DM the density is not obtained directly from the wavefunctions, so we can choose our initial density.
- We choose to start with the sum of the atomic densities. This makes the initial Hamiltonian pretty good compared to the random start EDFT has.

EDFT output (iprint: 1)

SCF loop	Energy	Fermi energy	Energy gain per atom	Timer (sec)	<-- SCF
Initial	5.22894717E+003	1.23054649E+002		23.25	<-- SCF
1	-1.34061398E+003	1.54028093E+002	4.69254368E+002	406.21	<-- SCF
2	-6.24970545E+003	3.25481388E+001	3.50649391E+002	773.44	<-- SCF
3	-6.44753749E+003	1.58875589E+001	1.41308601E+001	1122.97	<-- SCF
4	-6.48366756E+003	8.21631220E+000	2.58071892E+000	1478.40	<-- SCF
5	-6.48822994E+003	6.28831601E+000	3.25884507E-001	1829.99	<-- SCF
6	-6.48935120E+003	5.38297421E+000	8.00902664E-002	2175.23	<-- SCF
7	-6.48970423E+003	5.09069781E+000	2.52159741E-002	2529.80	<-- SCF
8	-6.49027213E+003	4.60235674E+000	4.05645475E-002	2892.63	<-- SCF
9	-6.49049787E+003	4.46602537E+000	1.61242962E-002	3252.49	<-- SCF
10	-6.49069177E+003	4.32576189E+000	1.38501648E-002	3609.24	<-- SCF
11	-6.49082746E+003	4.10031919E+000	9.69195382E-003	3981.20	<-- SCF

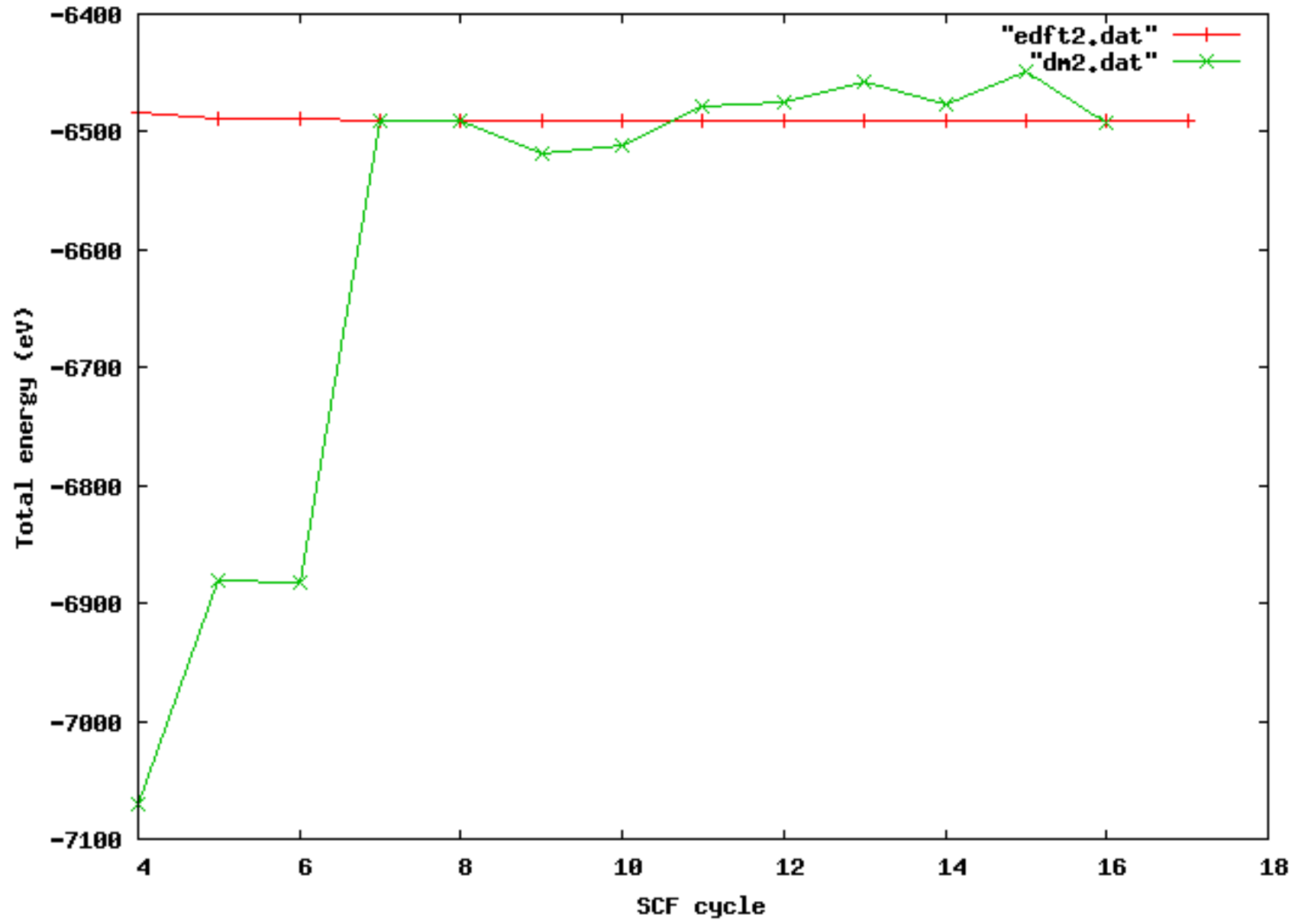
DM output (iprint: 1)

SCF loop	Energy	Fermi energy	Energy gain per atom	Timer (sec)	<-- SCF
Initial	-5.14317516E+003	5.12276157E+001		12.03	<-- SCF
1	-6.31741967E+003	6.63244417E+000	8.38746075E+001	20.60	<-- SCF
2	-6.49931311E+003	3.75941681E+000	1.29923889E+001	29.84	<-- SCF
3	-6.51038834E+003	3.54231581E+000	7.91087819E-001	39.34	<-- SCF
4	-7.06998837E+003	2.22343892E+000	3.99714308E+001	55.71	<-- SCF
5	-6.88101641E+003	3.72113246E+000	-1.34979974E+001	64.26	<-- SCF
6	-6.88234462E+003	3.67678710E+000	9.48725740E-002	73.81	<-- SCF
7	-6.48977868E+003	3.73752136E+000	-2.80404247E+001	84.70	<-- SCF
8	-6.49044610E+003	3.74216565E+000	4.76729934E-002	93.59	<-- SCF
9	-6.51835347E+003	4.26880418E+000	1.99338358E+000	107.69	<-- SCF
10	-6.51118474E+003	4.28640804E+000	-5.12052231E-001	120.24	<-- SCF
11	-6.47752722E+003	3.96827211E+000	-2.40410830E+000	133.02	<-- SCF
12	-6.47436227E+003	4.17156369E+000	-2.26067919E-001	144.39	<-- SCF
13	-6.45824918E+003	4.14265903E+000	-1.15093520E+000	155.99	<-- SCF
14	-6.47675286E+003	4.55171279E+000	1.32169139E+000	167.02	<-- SCF
15	-6.44884485E+003	4.41836929E+000	-1.99342920E+000	178.41	<-- SCF
16	-6.49295003E+003	4.31979139E+000	3.15036998E+000	191.27	<-- SCF
17	-6.49378104E+003	4.31843062E+000	5.93581671E-002	201.07	<-- SCF
18	-6.44552085E+003	4.71784691E+000	-3.44715632E+000	213.32	<-- SCF
19	-6.44589421E+003	4.73581949E+000	2.66682897E-002	222.73	<-- SCF
20	-6.41032124E+003	5.23541858E+000	-2.54092638E+000	236.73	<-- SCF

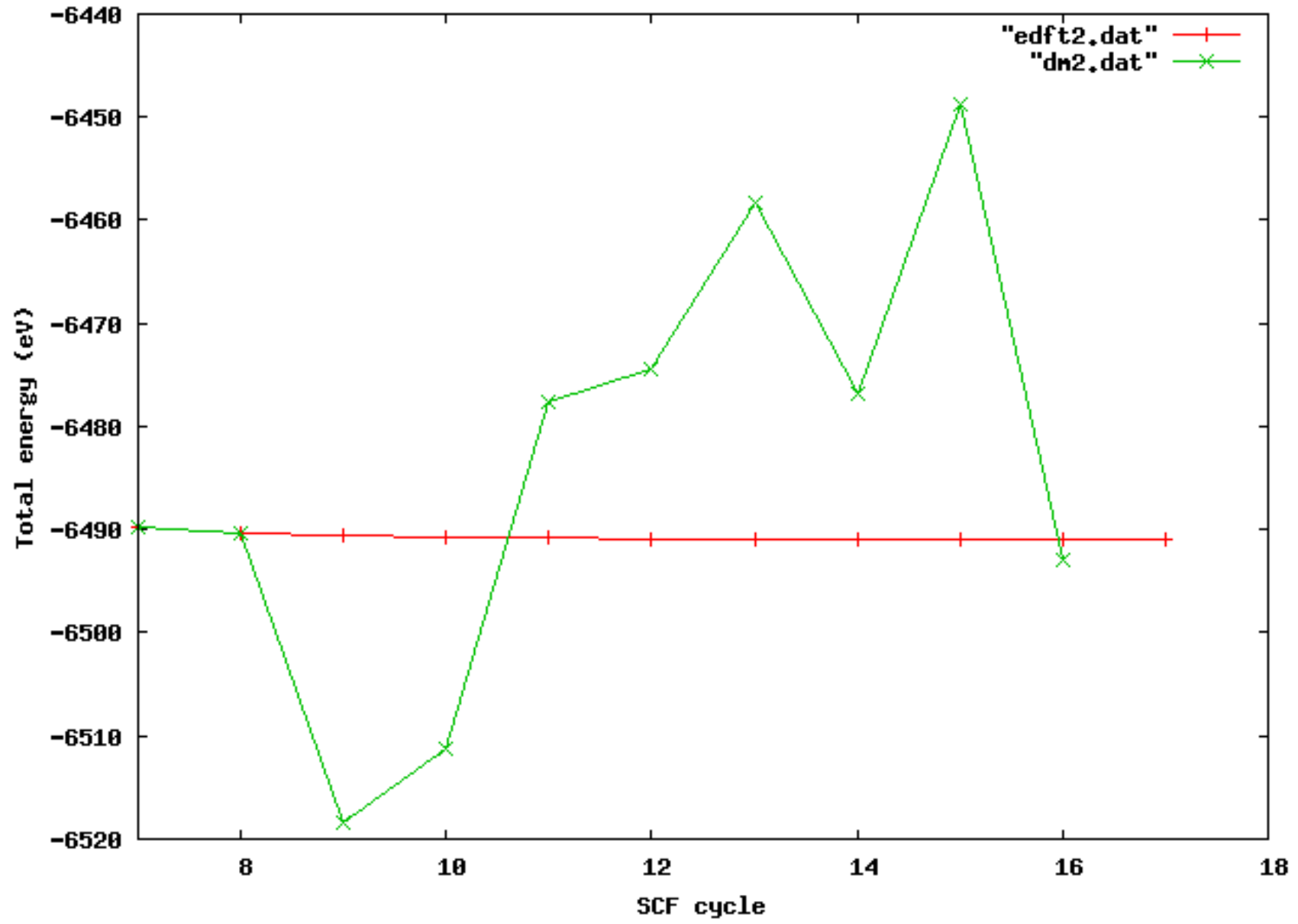
EDFT vs DM (Part II)

- Test system: erbium and hydrogen in silicon (340eV)
- EDFT converges in 14 SCF cycles (4000s)
- DM doesn't converge!
- EDFT is the clear winner.

Total energy convergence for EDFT and DM



Total energy convergence for EDFT and DM



DM's Problems

- What's going wrong with DM?
- The larger the system, the smaller the smallest nonzero wavevector is, so sloshing is worse.
- Sloshing is made worse if the material has a large dielectric susceptibility, so metals can be especially problematic.
- Spin can also make matters worse, because there may be many metastable spin states, and the system can now **spin slosh** as well as charge slosh.

Castep scaling

- Constructing the density and applying the local potential require FFTs of each band from reciprocal to real-space. FFTs scale as $N \log N$.
 - In **G-vector parallel** calculations the prefactor **increases** with the number of processors.
- Orthonormalisation scales as N^3 .
 - In parallel calculations the prefactor **decreases** with the number of processors.
- Parallel calculations scale almost perfectly when pure **k-point parallelisation** is used.

Summary

- EDFT is stable and fully variational, giving excellent forces and stresses; but it is slow.
- DM is fast, but non-variational. Since the density and wavefunctions are not quite consistent, better convergence tolerances are needed to get accurate forces and stresses.
- DM is usually the fastest, but can be unstable when there are large systems, band crossings, multiple spin states, and/or the dielectric susceptibility is large.